# Phase-field simulations of solidification structures

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## **Introduction: Real Structures**

#### Experimental solidification structures: Snowflakes and Frost





http://www.its.caltech.edu/~atomic/snowcrystals/



# **Real Structures**

Experimental solidification structures: Dendrites in transparent materials





M. E. Glicksman organic material

J. H. Bilgram, ETH Zürich Xenon





interface thickness (MD)  $\sim 10 \text{\AA}$ 

W. J. Boettinger, J. A. Warren, C. Beckermann, and A. Karma. Annu. Rev. Mater. Res. 32 (2002) 163



- entropy functional

$$S(e,c,arphi) = \int \left[ s(e,c,arphi) - \left( arepsilon a(
abla arphi) + rac{1}{arepsilon} w(arphi) 
ight) 
ight] dx,$$



 $a(
abla arphi) = \gamma(
abla arphi)^2, \hspace{0.3cm} w(arphi) = 9\gamma g(arphi), \hspace{0.3cm} g(arphi) = arphi^2(1-arphi)^2.$ 



- evolution of phase field

$$\partial_t arphi \sim rac{\delta S}{\delta arphi},$$

$$2\omegaarepsilon\partial_tarphi=2arepsilon\gamma
abla^2arphi-rac{9\gamma}{arepsilon}g_{,arphi}(arphi)-rac{1}{T}f_{,arphi}(c,arphi),$$

phase-field parameters:  $\varepsilon, \gamma, \omega$ .

- free energy density (T = const)

$$f(c, arphi) = \sum_{i=1}^{2} c_i L_i rac{T - T_i}{T_i} h(arphi) + rac{RT}{v_m} \sum_{i=1}^{2} c_i \ln c_i,$$
 $h(arphi) = arphi^2 (3 - 2arphi).$ 



- chemical potential

$$\mu_i = f_{,c_i} = L_i \frac{T - T_i}{T_i} h(\varphi) + \frac{RT}{v_m} (\ln c_i + 1),$$

mass fluxes

$$egin{aligned} &J_i = \sum\limits_{j=1}^2 L_{ij} 
abla rac{-\mu_j}{T}, \ &L_{ij}(c,arphi) = rac{v_m}{R} D_i c_i \left( \delta_{ij} - rac{D_j c_j}{\sum\limits_{k=1}^2 D_k c_k} 
ight), \end{aligned}$$

 $c_1 + c_2 = 1.$ 

H. Garcke, B. Nestler, B. Stinner, *A diffuse interface model for alloys with multiple components and phases*, SIAM J. Appl. Math. **64**(3) (2004) 775–799.



# **Binary alloy**



Equilibrium phase diagram



# **Binary alloy: Evolution equations**

phase field (nonconserved field)

$$\frac{\partial \varphi}{\partial t} = \frac{\varepsilon^2}{L^2} \nabla^2 \varphi - \frac{9}{2} \frac{dg}{d\varphi} - \frac{\varepsilon}{2\gamma} \Lambda(c, u) \frac{dh}{d\varphi}$$

$$\Lambda(c,u) = (1 - c_0 c) \frac{L_A}{T_A} \frac{u + (T_0 - T_A)/T_Q}{u + T_0/T_Q} + c_0 c \frac{L_B}{T_B} \frac{u + (T_0 - T_B)/T_Q}{u + T_0/T_Q}$$

- concentration field (conserved field)

$$egin{aligned} rac{\partial c}{\partial t} &= 
abla ig(ar{D}
abla cig) - 
abla ig(ar{D}c(1-c_0c)
abla (\Theta(u)h(arphi))ig) \ &ar{D} &= rac{D}{
u}rac{arepsilon^2}{L^2} \ &egin{aligned} \Theta(u) &= rac{v_m}{R} \left(rac{L_A}{T_A}rac{u+(T_0-T_A)/T_Q}{u+T_0/T_Q} - rac{L_B}{T_B}rac{u+(T_0-T_B)/T_Q}{u+T_0/T_Q} 
ight) \end{aligned}$$



# **Different length scales**





# Adaptive grid spacing



$$egin{aligned} E &= \sum_lpha |
abla \phi_lpha| d^2 \ &+ E_C \sum_j |
abla c_j| d^2 \ &+ E_T |
abla T| d^2. \end{aligned}$$

The cell will be refined if  $E > E_{max}$  and coarsened if  $E < E_{min}$ . By this choice,  $d \sim 1/\sqrt{E}$ .



# **FEM:** phase field and concentration field





phase field

concentration field



# **FEM: Adaptive grid**





# **Anisotropy of the interface**

gradient energy  $a(\nabla \varphi) = \gamma (1 + \delta_{\gamma} \cos \beta) (\nabla \varphi)^2$ 



$$\delta_\gamma=0$$

 $\delta_{\gamma}=0.04$ 



#### Noise

#### driving force ightarrow (1+lpha r) imes driving force, -1 < r < +1



$$\delta_{\gamma}=0,\, lpha=0.25$$

$$\delta_{\gamma}=0.04,\,lpha=0.25$$

















4600 time steps

 $1303 \rightarrow 347992$ active cells  $\frac{L}{----=256}$ 



- Periodic boundary conditions.
- Output to binary OpenDX-files.
- Platform-independent binary format in Vector<Number>::block\_write/block\_read.

